

## FABRICATION AND UTILIZATION OF ZINC OXIDE NANOFIBRES FOR SEMICONDUCTORS OF PHOTOVOLTAIC- AND PIEZOELECTRIC-BASED HYBRID GENERATORS

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### Abstract

The objective of this study is to develop and test hybrid generators harvesting both solar and mechanical energy. The hybrid generators work using photovoltaic- and piezoelectric-principles, where the active semiconductors were ZnO nanofibres synthesized by electrospinning at various feed rates of precursor. The results show that the crystallinity, crystalline size, fibre size, and fibre uniformity examined by X-ray diffraction and scanning electron microscopy were strongly affected by changing the solution flow rate among 2, 4, 6, and 8  $\mu\text{L}/\text{min}$ . Meanwhile, the voltage and power generated from the hybrid generators were influenced by the crystalline quality and morphology of the ZnO fibres. The highest voltages of photovoltaic- and piezoelectric-based hybrid generators were 507 and 36.3 mV, respectively. In addition, the maximum power values of photovoltaic- and piezoelectric-based generators were 706.72  $\mu\text{W}/\text{cm}^2$  and 49.6  $\text{nW}/\text{cm}^2$ , respectively. Therefore, hybrid generators are of interest for fabrication into self-power devices.

Keywords: Nanofibre, Semiconductors, Piezoelectric, Photovoltaic, Hybrid nanogenerator.

### 1. Introduction

Small devices for harvesting renewable energy are of much interest for personal electronics, personal computers, and sensors. Currently, such devices are driven by batteries or capacitors requiring periodic charging from external energy. Self-

**Nomenclatures**

$A$	Active area, m <sup>2</sup>
$D$	Size of the crystal
$G$	Irradiance intensity, W/m <sup>2</sup>
$h$	Planck's constant, $6.626 \cdot 10^{-34}$ Js
$I_{mpp}$	Current at the maximum power point, A
$k$	A constant, 0.9
$P_L$	Output power, Watt
$P_{max}$	Maximum output power, Watt
$R_L$	Electric load, $\Omega$
$T$	Period, s
$V_{mpp}$	Voltage at maximum power point, Volt
$V_o$	Output voltage, Volt
$\nu$	Frequency

**Greek Symbols**

$\alpha$	Absorbance
$\beta$	Full width at half maximum intensity
$\eta$	Efficiency of the solar cell
$\theta$	Diffraction angle
$\lambda$	Wavelength

**Abbreviations**

CoAc	Cobalt Acetate
DSSC	Dye-Sensitized Solar Cell
FTO	Fluorine-doped Tin Oxide
FWHM	Full Width at Half-Maximum
NG	Nanogenerator
PVA	Polyvinyl Alcohol
PVDF	Polyvinylidene Fluoride
PVP	Polyvinyl Pyrrolidone
PZT	Lead Zirconate Titanate
SEM	Scanning Electron Microscopy
TiO <sub>2</sub>	Titanium Dioxide
XRD	X-Ray Diffraction
ZnAc	Zinc Acetate
ZnO	Zinc Oxide

powered nanogenerators (NGs) have been developed not only using a single energy source [1-10] but also using various energy sources [11-13]. Therefore, one of the key issues is how to harvest electricity from multiple renewable energy sources in a single and compact nanogenerator device.

NGs are mainly fabricated from semiconductors. TiO<sub>2</sub> has been intensively studied for dye-sensitized solar cells (DSSCs) because it produces superior performance compared to that produced by ZnO-based DSSCs. In contrast, the uniqueness of non-centrosymmetric ZnO crystals may produce a piezoelectric effect. To date, lead-free materials such as ZnO [14], 0.755(Bi<sub>0.5</sub>Na<sub>0.5</sub>)TiO<sub>3</sub>-0.065BaTiO<sub>3</sub>-0.18SrTiO<sub>3</sub> (BNT-BT-ST) [15], polyvinylidene fluoride (PVDF) [16],

and  $(K_{0.5}Na_{0.5})NbO_3$  [17] have been intensively developed for piezoelectric materials. In addition, ZnO, with some other unique properties including a wide energy band gap of 3.61 eV [18], intrinsic n-type semiconductivity, and resistance to high temperatures, has been used for the construction of various equipment, e.g., solar cells [19-21], gas sensors [22, 23], and piezoelectric-based devices [24-28]. Unfortunately, the ability of ZnO to act as a semiconductor in energy conversion devices is still limited to a single source. Thus, the study of ZnO in hybrid devices proves to be an interesting topic.

ZnO-based NGs have been studied using nanowires [8], nanorods [14], and nanotubes [9] as a single energy source. Moreover, hybrid NGs have also been developed using CdSe combining piezoelectric and photoelectric concepts [12], using nanopillars for solar cells and PVDF for the piezoelectric effect [29], using piezoelectric and triboelectric effects [30], and using ZnO nanosheets for organic photovoltaic and piezoelectric NGs [31]. To the best of our knowledge, although different types of hybrid devices have been developed, hybrid NGs for harvesting solar energy using the concept of dye-sensitized solar cells (DSSCs) and mechanical energy using the piezoelectric effect have not been reported, except from Kim et al. [32]. The hybrid NGs used flexible DSSCs, whereas the semiconductors used were ZnO nanorods. Meanwhile, the piezoelectric effect was accomplished by bending loads. An efficiency of 0.69% with  $J_{sc}$  (3.95 mA/cm<sup>2</sup>) and  $V_{oc}$  (0.65 V) were produced when hybrid NGs worked as DSSCs [32]. In addition, an output voltage of approximately 1.5 V with an output current density approaching 80 nA/cm<sup>2</sup> was generated when hybrid NGs worked using the piezoelectric effect [32]. Unfortunately, the main type of mechanical load on the keyboards is a compressive load instead of a bending load.

Therefore, the objective of the study is to demonstrate and investigate the hybrid NGs for harvesting solar energy using the concept of DSSCs and mechanical energy using the piezoelectric effect, where the semiconductors used were ZnO nanofibres and the mechanical load used was a cyclic compressive load. For better understanding, the effects of the precursor feed rate in the electrospinning on the morphology, crystallite properties, and performance of hybrid NGs were also studied. The hybrid NGs have advantages in their design because they have just one active area and are therefore simple to manufacture. The fibres of ZnO fabricated by an electrospinning machine [33, 34] were attached onto conductive transparent glass by a screen-printing method.

## 2. Experimental Procedures

### 2.1. Materials

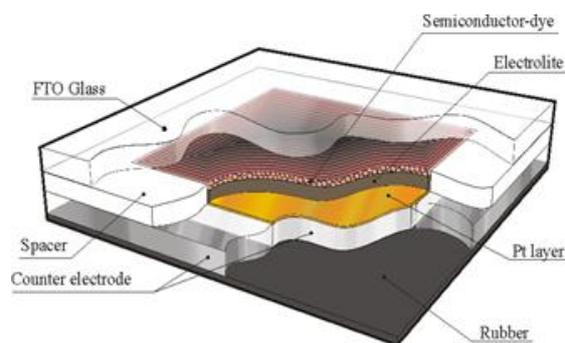
For synthesizing the fibres of ZnO, we used purified water, polyvinyl alcohol ( $[-CH_2CHOH-]_n$ , PVA 72000, Merck), zinc acetate dihydrate ( $Zn(C_2H_3O_2)_2 \cdot 2H_2O$ , ZnAc, Merck), and cobalt(II) acetate tetrahydrate ( $(CH_3COO)_2Co \cdot 4H_2O$ , CoAc, Merck). The materials were used as received. The precursor solution in the electrospinning machine is PVA-based with a molecular weight of 72,000, and it has a good enough viscosity and conductivity to form a Taylor cone and create fibres by using an electrostatic field.

## 2.2. Fabricating and testing the hybrid NGs

The hybrid NGs were fabricated from FTO (fluorine-doped tin oxide) glass, ZnO fibres attached onto the FTO glass, N719 dye solution, electrolytes, counter electrode glass, and a rubber sheet. The hybrids NGs, as shown in Fig. 1, were assembled as DSSCs, where the counter electrode was attached onto the rubber sheet as a flexible mounting. The counter electrodes could freely move upward and downward when the FTO glass was subjected loads.

FTO (fluorine-doped tin oxide) glass was synthesized by a previously reported method [35]. Meanwhile, the counter electrodes were made with the base substrate using FTO, where platinum was sputtered and deposited onto FTO. Platinum was placed in the vacuum room with a base pressure of  $9.5 \times 10^{-5}$  Torr, while argon gas was maintained at 4 mTorr before entering the vacuum space. The applied potential and current were 404 volts and 125 mA, respectively. The substrate was rotated at a speed of 5 rpm to produce a uniform deposition. The sputtering process for producing the counter electrode was carried out for 20 minutes. In addition, the preparation of the electrolyte and N719 dyes followed previous procedures [36].

To fabricate the hybrid NGs, the FTO glass with ZnO fibres was dipped in the N719 dye solution for 24 hours. Next, the FTO and counter electrode glass were attached and given a spacer with a distance of approximately 35  $\mu\text{m}$ . Electrolytes were incorporated into the gap between the two glasses and then sealing was performed. The counter electrode glass was attached on the pliable rubber so that the ZnO semiconductor could be activated by applying the mechanical load. The order of layers of the hybrid NGs is shown in Fig. 1.



**Fig. 1. Scheme of the hybrid NGs.**

The ability of hybrid NGs to produce electrical energy by means of the piezoelectric effect was measured by an ADAM Advantech (series 4018) instrument, and the output voltage was read in real-time with the application of a compressive load at 0.5 kg. The load of 0.5 kg was applied to represent the condition of weak finger taps [37]. The power of piezoelectric-based NGs was then calculated using Formula 1 from the output voltage generated by providing the variation of the load resistance ( $\Omega$ ) [38].

$$P_L = \frac{1}{T} \int \frac{V_o(t)^2}{R_L} dt \quad (1)$$

The ability of hybrid NGs to produce electrical energy by means of the photovoltaic effect was measured by a solar simulator at an irradiance of 1000 W/m<sup>2</sup>. The power of photovoltaic-based NGs was then calculated using Formula 2 from the point of maximum power ( $mpp$ ,  $P_{max}$ ). Meanwhile, Formula 3 calculated the efficiency of the hybrid NG based on the photovoltaic effect.

$$P_{max} = V_{mpp}I_{mpp} \quad (2)$$

$$\eta = \frac{P_{max}}{A \times G} \quad (3)$$

### 2.3. Synthesis and characterization of ZnO fibres

The ZnO fibres used for the photo anode in photovoltaic-based NGs as well as for the active layer in piezoelectric-based NGs were attached onto the FTO glass. The ZnO fibres were synthesized using electrospinning with a solution of zinc acetate (ZnAc) and poly vinyl alcohol (PVA). To synthesize the PVA solution, the PVA was dissolved in purified water at a mass ratio of 1:10. The PVA (2.0 g) and 20 mL of purified water were mixed and stirred until homogeneous at a temperature of 70°C for 4 hours, as mentioned previously [39]. The bubbles and foam were relieved for 24 hours at a temperature of 25°C.

To synthesize the solution of ZnAc, 2 g of ZnAc was mixed with 8 mL of purified water, stirred at a temperature of 70°C for 1 hour, and allowed to settle for 24 hours at a temperature of 25°C. The ZnAc/PVA solution was synthesized by mixing the ZnAc and PVA solutions at a weight ratio of 4:1. The last solution was mixed at a temperature of 70°C for 8 hours, followed by settling for 24 hours at a temperature of 25°C.

Moreover, 1.0 mL of ZnAc/PVA solution was put in the syringe pump, where the flow rates of the syringe pump in the horizontal electrospinning machine were 2, 4, 6, and 8  $\mu$ L/min, respectively, and each solution was electrospun to produce the green ZnAc/PVA fibres. The collector of the electrospinning machine is FTO (fluorine-doped tin oxide) glass synthesized by a previously described method [35].

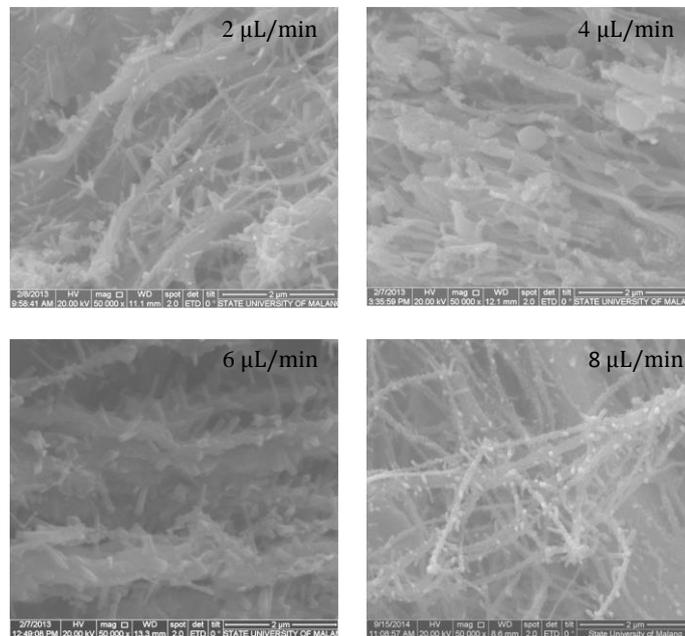
In addition, sintering of the green fibres was conducted at a temperature of 550°C for 2 hours to remove PVA because above 440°C, it was completely decomposed [40] and all that remained was the ZnO nanofibres. The morphology was characterized and crystallinity of the nanofibres was studied by scanning electron microscopy (SEM, FEI: Inspect-S50) and X-ray diffraction (XRD).

## 3. Results and Discussion

### 3.1. Morphology and crystallinity

Figure 2 shows SEM photographs of ZnO fibres synthesized by the electrospinning machine at various flow rates of precursor solution. It can be seen that the fibres are dense, long, absent of beads, rough, and have wires on their surface. Evidently, the long and dense fibres show that the ZnAc was well-dispersed in the PVA matrix and acted as nuclei in the conversion to ZnO crystals. The growth of ZnO crystals is localized, limited, and might coalesce within the PVA matrix [41]. Moreover, the formation of free beaded fibres means that the applied voltage and the precursor feed rate are sufficient to stretch the filaments from the needle tip to the FTO collector.

However, the produced fibres are different when compared with a previous study, where their fibres were grainy with the use of a precursor solution from ZnAc/PVP [33]. PVA is an atactic polymer [42], and therefore, it crystallises only to a very small extent because it lacks any regularity in the arrangement of substituents. Meanwhile, PVP contains almost equal proportions of isotactic and syndiotactic configurations [43]. In the syndiotactic configuration, the substituents have alternate positions along the chain. Meanwhile, in the isotactic configuration, all of the substituents are located on the same side of the macromolecular backbone. Due to their random nature, atactic PVA is usually amorphous. In contrast, isotactic and syndiotactic PVP are usually semi-crystalline. PVP is also a polymer possessing activity as a shape-control agent in the synthesis of nanocrystals [44, 45], promoting the growth of specific crystal faces while suppressing the growth of others. In addition, the syndiotactic PVP also has a rapid rate of crystallization. Thus, the fibres synthesized by the electrospinning of ZnAc/PVP solution are grainy. Meanwhile, ZnO fibres synthesized using ZnAc/PVA precursors tend to be amorphous and covered by wires. This is also confirmed by the XRD spectra, in which high intensities occurred in the (101) and (100) crystal planes with moderate crystallinity.



**Fig. 2. SEM images of ZnO fibres at various flow rates of precursor solution into the electrospinning machine.**

The diameter of the fibres is strongly affected by the precursor feed rate into the electrospinning machine [46]. Empirical observations indicate that the smallest diameters occur at the lowest flow rates, but the production rate and fibre size tend to vary with the solution concentration [47]. Increasing the precursor flow rate and decreasing the applied voltage tend to increase the fibre diameter and bead diameter [47] because the electric field cannot overcome the surface tension of precursors. By increasing the flow rate for the same electric field, the

average diameter of nanofibres increased due to the decrease of the charge in the unit volume of the precursor liquid [47, 48]. In addition, the diameter of ZnO fibres is also more homogeneous at a lower flow rate of precursor solution, as indicated by a smaller standard deviation and shown in Table 1. The smallest mean diameter of ZnO fibres is 142.7 nm, and it was obtained at a flow rate of 2  $\mu\text{L}/\text{min}$ . It is important to note that the smallest flow rate is 2  $\mu\text{L}/\text{min}$  because Taylor's cone and good fibres were very difficult to synthesize with the electrospinning machine with a needle diameter of 0.8 mm, applied voltage of approximately 15 kV, and solution flow rate below 2  $\mu\text{L}/\text{min}$ .

The diameters of ZnO fibres ranged from 142.7-442.3 nm. In contrast, the fibre diameter found in this study is slightly larger than those which resulted from previous studies, with values ranging from 65.3 to 82.8 nm at the same precursor flow rate [39]. The cause is that the collector used in the previous studies is an aluminium plate with a greater electrical conductivity than that of the FTO glass. The greater the electrical conductivity, the higher the speed the solution precursor was drawn for producing smaller diameter ZnO fibres.

**Table 1. Diameter of ZnO fibres.**

Size	Flow rate of precursors			
	2 $\mu\text{L}/\text{min}$	4 $\mu\text{L}/\text{min}$	6 $\mu\text{L}/\text{min}$	8 $\mu\text{L}/\text{min}$
<b>Mean diameter</b>	142.7	180.3	432.0	442.3
<b>Deviation standard</b>	43.2	57.1	90.5	163.9

Meanwhile, X-ray diffraction spectra of ZnO fibres are shown in Fig. 3. The three main diffraction peaks have indices of (100), (002), and (101) corresponding to the hexagonal structure of ZnO crystals based on JC-PDF 36-1451. The increase in the flow rate of the precursor solution in the synthesis of ZnO fibres with an electrospinning machine does not change the X-ray diffraction pattern, revealing that no new phase was formed. As the precursor flow rates increased, the intensity of the diffraction peaks increased and narrowed the full width at half maximum (FWHM).

Furthermore, the crystalline diameter of ZnO was calculated using the Debye-Scherrer's formula at the highest peak [49]:

$$D = \frac{k\lambda}{\beta \cos\theta} \quad (4)$$

Meanwhile, Williamson and Hall explained that the strain ( $\epsilon$ ) in the crystal has an effect on the changes in the crystalline size. According to Williamson-Hall equations, crystalline size and strain relationships can be expressed by [50, 51]:

$$\beta \cos\theta = \frac{k\lambda}{D} + 4\epsilon \sin\theta \quad (5)$$

The crystalline sizes of ZnO fibres were calculated with Eqs. (1) and (2) and are shown in Table 2. Meanwhile, the crystallinity of ZnO fibres ranged between 71.07 and 74.28%. In the synthesis of ZnO fibres with the electrospinning machine, the crystalline size was also influenced by the flow rate of precursor solution. The greatest crystalline size of the ZnO fibres occurred at a flow rate of 4  $\mu\text{L}/\text{min}$ . The crystalline size of the ZnO fibres also increased because of internal strain. The greatest strain on the ZnO fibres was 0.315% when the flow rate of precursor solution was 4  $\mu\text{L}/\text{min}$ .

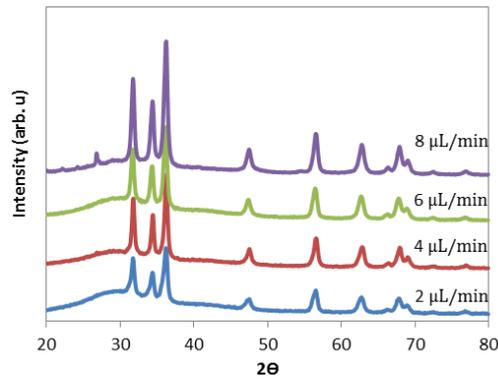


Fig. 3. X-ray diffraction spectra of ZnO fibres.

Table 2. Band gap energy, crystalline size, and crystallinity of ZnO fibres.

Flow rate ( $\mu\text{L}/\text{min}$ )	Band gap energy (eV)	Crystallinity (%)	Crystalline size (nm)		Strain (%)
			Debye-Scherrer	Williamson-Hall	
2	3.17	73.67	13.9	22.2	0.207
4	3.19	74.28	17.6	45.1	0.315
6	3.22	73.24	15.6	34.6	0.297
8	3.29	71.07	15.5	27.9	0.222

The UV-vis absorption spectra are shown in Fig. 4. The band gap energy of ZnO fibres was calculated by the method of the Tauc plot [52, 53]. The optical band gap energy ( $E_g$ ) of ZnO fibres was determined from the intercepts with the photon energy (x-axis) by linear fitting of the  $(\alpha \cdot hv)^2$  vs.  $hv$  curve, as shown in Fig. 4 [54, 55]. The band gap energy of ZnO fibres calculated by the method of the Tauc plot ranges from 3.17 to 3.29 eV. Similar results can also be observed in some previous studies [55]. Therefore, the flow rate of precursor solution led to an increase in the band gap energy of synthesized ZnO fibres.

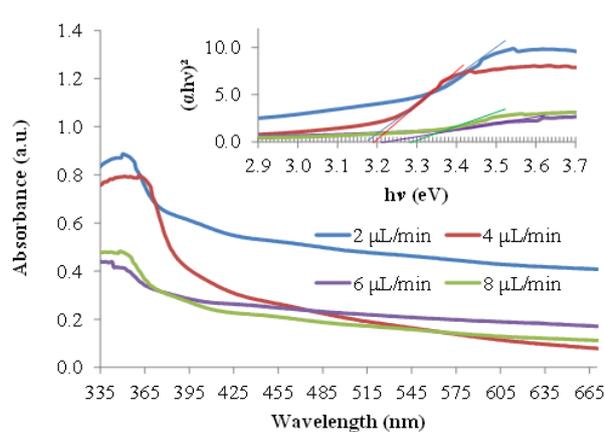


Fig. 4. UV-Vis absorption spectra of ZnO fibres; insert is the curve of  $(\alpha hv)^2$  versus photon energy ( $h\nu$ ).

### 3.2. Performance of hybrid NGs (NGs)

A schematic of the hybrid NGs in this research can be seen in Fig. 1. The hybrid NGs were the solar cell of DSSCs, where part of the counter electrode is freely moving with the elasticity obtained from rubber attached onto the back electrode. When exposed to light, the hybrid NGs work as solar cells, and when subjected to the mechanical load, the cells serve as piezoelectric-based NGs.

The characteristics of the current and voltage of the hybrid NGs when exposed to a solar simulator irradiance of  $1000 \text{ W/m}^2$  are shown in Fig. 5. Meanwhile, the performance of the hybrid NGs when subjected to a load of  $0.5 \text{ kg}$  is shown in Fig. 6.

When the hybrid NGs work as photovoltaics, the open circuit voltage ( $V_{oc}$ ) generated ranges from  $0.421$  to  $0.507 \text{ V}$ . As the flow rate of the solution precursors during the process of electrospinning increases, the generated  $V_{oc}$  by NGs experiences a slight decline because of the wider band gap energy from the semiconductor made of ZnO fibres. Furthermore, the  $J_{sc}$  of photovoltaic-based hybrid NGs decreased quite sharply with an increasing flow rate of precursor solution during the electrospinning process. This is caused by the size of the ZnO particles. The bigger the size of the fibres, the weaker the bond formed between the semiconductor particles and the dye molecules. One of the main factors influencing  $J_{sc}$  is the anchoring or bonding between the semiconductor particles and dye molecules [36, 56]. Photovoltaic-based hybrid NGs generated the highest  $J_{sc}$  of  $0.993 \text{ mA/cm}^2$ .

In addition, besides the size of fibres, the uniformity of ZnO fibres may also determine the level of  $J_{sc}$ . Uniform fibres accelerate the flow of electrons. Furthermore, the uniformity of fibres synthesized by the electrospinning machine has become a major issue apart from the size of fibres [57]. The fill factor (FF) and efficiency of photovoltaic-based hybrid NGs are affected by the flow rate of the precursor solution. The highest efficiency of photovoltaic-based hybrid NGs of  $0.24\%$  was produced by a flow rate of  $2 \mu\text{L/min}$  and influenced by the grain size of  $\text{SnO}_2:\text{F}$  in the FTO glass of  $470 \text{ nm}$  [35]. Better contact may occur between the grains of  $\text{SnO}_2:\text{F}$  ( $470 \text{ nm}$ ) and the fibres of ZnO ( $142.7 \text{ nm}$ ) produced by a flow rate of  $2 \mu\text{L/min}$ . The appropriate relationship between the grain size of  $\text{SnO}_2:\text{F}$  and the fibre size of ZnO provides good and strong bonding, thus enhancing the transfer of electrons.

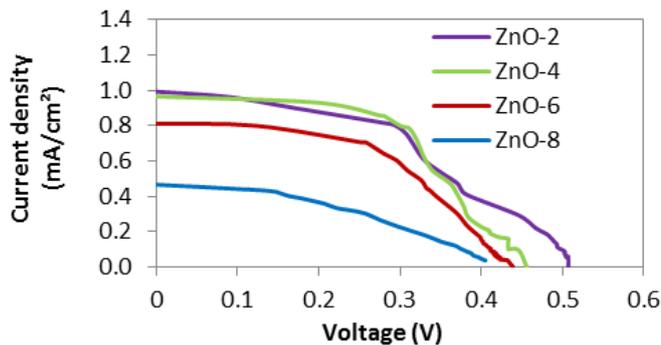
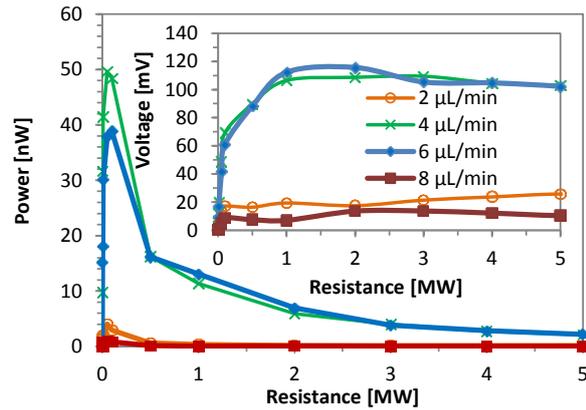


Fig. 5. I-V curves of photovoltaic-based hybrid NGs.



**Fig. 6. Output power of piezoelectric-based hybrid NGs.**

Meanwhile, the highest output power from the ZnO fibre-based hybrid NGs was also higher than that of the lead zirconate titanate (PZT) fibre-based NGs. The highest output power of the lead zirconate titanate (PZT) fibre-based NGs was 30 nW at 6 MΩ [38].

Next, the NGs were tested for their ability to produce electricity from the mechanical load as shown in Fig. 5. The highest output voltage and power were 352 mV and 48.2 nW, respectively, when the ZnO fibres were synthesized by an electrospinning machine at a flow rate of 4 μL/min. The output voltage of ZnO fibre-based hybrid NGs was higher than that of ZnO-PSS/PVA thin layer-based NGs (varying from 10 to 100 mV) [58].

Moreover, the output voltage and power are strongly affected by the positions of Zn and O atoms in a hexagonal structure of ZnO because the dipole moment is generated from the difference in the electronegativity and positions of these two atoms. The ZnO fibres in the hybrid NGs exhibited internal strains in their crystal structure influencing the increase of their size compared to the size of the crystal without strain, as seen in Table 2. The existence of a greater strain stretched apart the positions of both of the atoms, and therefore, the output voltage increased.

Generally, the highest power generated from photovoltaic-based hybrid NGs is still far greater than that generated from piezoelectric-based hybrid NGs. The highest powers generated from photovoltaic- and piezoelectric-based hybrid NGs are 706.72 μW and 49.6 nW, respectively, as shown in Fig. 6. Therefore, the power densities of 706.72 μW/cm<sup>2</sup> and 49.6 nW/cm<sup>2</sup> are very promising for self-powered devices based on hybrid NGs. However, the power density of piezoelectric-based hybrid NGs needs to be enhanced by means of doping the ZnO fibres with suitable materials.

#### 4. Conclusions

Hybrid NGs were successfully fabricated and investigated. Hybrid-NGs work by using photovoltaic and piezoelectric principles. Piezoelectric-based NGs work when subjected to mechanical loads, while photovoltaic-based NGs work upon

receiving light irradiation. The semiconductor used was ZnO fibres synthesized by an electrospinning machine with various flow rates of precursor solution from 2 to 8  $\mu\text{L}/\text{min}$ . The crystallinity, crystalline size, fibre size, and fibre uniformity were strongly affected by the flow rate of the precursor solution. Meanwhile, the voltage and power generated from hybrid NGs were influenced by the crystalline quality and morphology of the ZnO fibres. The voltage and power generated from photovoltaic-based hybrid NGs are still far higher when compared to those generated from the piezoelectric-based hybrid NGs. The highest voltages of photovoltaic- and piezoelectric-based NGs were 507 and 36.3 mV, respectively. Meanwhile, the maximum powers of photovoltaic- and piezoelectric-based NGs were 706.72  $\mu\text{W}$  and 49.6 nW, corresponding to power densities of 706.72  $\mu\text{W}/\text{cm}^2$  and 49.6  $\text{nW}/\text{cm}^2$ , respectively. To enhance the performance of hybrid NGs, the properties of the ZnO fibres need to be improved by doping with suitable dopant materials. By doing so, the hybrid-NGs become suitable for fabrication into devices of self-powered NGs.

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